

XPS, UPS and ESR studies on the interfacial interaction in Ni-ZrO₂ composite plating

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The interfacial interaction in Ni-ZrO₂ composite plating prepared by composite electrodeposition has been investigated. The results showed that no new phases were formed at the interfacial regions between matrix Ni and ZrO₂ particles, but the orbital interaction through the mutual overlap of the d orbits does exist in the interfacial regions between Ni atoms and Zr³⁺ ions. © 2000 Kluwer Academic Publishers

1. Introduction

Composite materials have been widely used and are playing more and more important roles. The properties of composite materials are greatly affected by their microstructure. Up to now, the knowledge about their inner structure is much limited [1–3]. The development of modern analytical methods such as XPS and UPS have made it possible to obtain more information about the build-up and microstructure. Ni-ZrO₂ composite platings (CP), comprised of matrix metal Ni and ZrO₂ particles, were prepared by composite electrodeposition (CE), and they were used as electrodes for hydrogen evolution reaction (HER). Those Ni-ZrO₂ composite electrodes showed very high electrocatalytic activity, which was many times higher than that of Ni electrode [4] although ZrO₂ itself does not display any HER ability because of its big forbidden band width ($E_g = 5.0$ eV) [5], all the experimental phenomena suggest that some kinds of interactions that benefit HER should exist between matrix Ni and ZrO₂ particles. These interactions have been systematically investigated, and the results are presented here.

2. Experimental

Ni-ZrO₂ CP was prepared by the method of CE, and the composition of the electrodeposition solution and the electrodeposition conditions are listed in Table I. Direct current was used in the electrodeposition process. Ni plating as a control was also prepared under the same conditions except that there were no ZrO₂ particles in the solution. The average diameter of ZrO₂ particles is about 2.3 μm and ZrO₂ content is 17.8% in the CP.

Morphology and structure were analyzed by HITACHI X-650 SEM and HITACHI H-700 TEM. The detailed procedures for XPS, UPS and ESR are described in reference [6].

3. Results and discussion

3.1. Results of SEM and TEM

As shown in Fig. 1, ZrO₂ particles are well-distributed in the matrix Ni, and the structure of CP is homoge-

neous. Pure ZrO₂ particles used in our experiment are composed of many smaller ZrO₂ monocrystals (Fig. 2), and an electron diffraction pattern is illustrated in Fig. 3. It was proved that the crystal structure of ZrO₂ particles is monoclinic. Observations and analyses have been carefully carried out in the interfacial regions to try to find out if any kind of interactions, which may lead to the formation of new compounds, exists between matrix Ni and ZrO₂ particles. TEM (Fig. 4) and X-ray results indicate that no new phases are formed in the interfacial regions, so the research was focused on the orbital interaction in the interfacial regions.

3.2. Results of XPS

The XPS spectrum of pure ZrO₂ powder shows the binding energies of 182.2 eV (Zr3d_{5/2}) and 184.6 eV (3d_{3/2}) (Fig. 5a), and it proves that the valence number of zirconium in ZrO₂ powder is +4 [7]. By comparing Fig. 5b with Fig. 5a, it is found that the shape of the spectrum in Ni-ZrO₂ CP (Fig. 5b) has been changed greatly. Curve fitting has been done according to the peak shape and half-peak width of the spectrum of ZrO₂ powder, and the fitting results (Fig. 5b) coincide well with the tested one. The zirconium peak in Ni-ZrO₂ CP is composed of two groups, and their binding energies are as follows: $E_{\text{I}}(3d_{5/2}) = 180.3$ eV, $E_{\text{I}}(3d_{3/2}) = 182.9$ eV, $E_{\text{II}}(3d_{5/2}) = 181.8$ eV, $E_{\text{II}}(3d_{3/2}) = 184.6$ eV. The binding energies of 3d_{5/2} electron for group I and group II have decreased by 1.9 eV and 0.4 eV respectively comparing with that of pure ZrO₂ powder, while the decrease in binding energy indicates the reduction of Zr⁴⁺. All these suggest that the valence numbers of zirconium for the two groups are less than +4. Considering that the reduction of Zr⁴⁺ in ZrO₂ particles can only take place during the process of CE, and that the reduction of Zr⁴⁺ \rightarrow Zr³⁺ would be much easier than that of Zr³⁺ \rightarrow Zr²⁺, and also that the Zr3d_{5/2} electron binding energy of metal zirconium is 178.5 eV [7], it is, therefore, believed that the valence number of zirconium corresponding to group I is +3. Since the decrease in binding energy for group II

TABLE I Solution and electrodepositing conditions

NiSO ₄ · 6H ₂ O	AR	290.1 (g/l)
NiCl ₂ · 6H ₂ O	AR	50 (g/l)
H ₃ BO ₃	AR	40 (g/l)
ZrO ₂ particles	AR	3–24 (g/l)
Temperature		40 °C
pH		2.8
Anode electrode		Ni plate

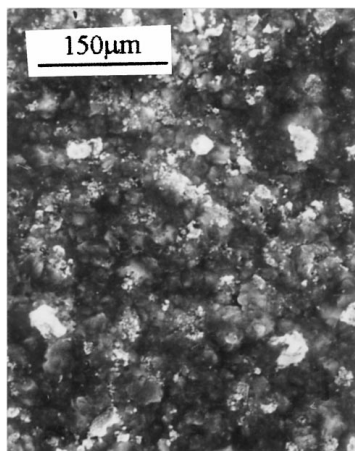


Figure 1 SEM photograph of Ni-ZrO₂ plating.

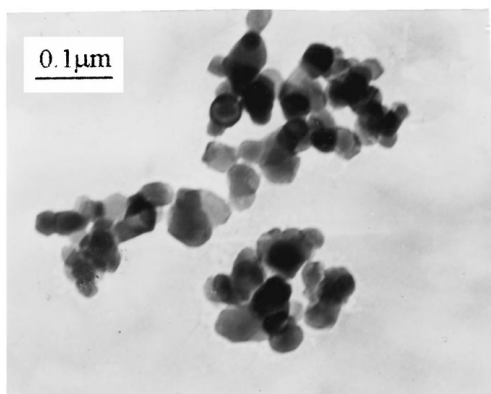


Figure 2 TEM photograph of ZrO₂ particles.

is less than 1 eV, it is thought that most of zirconium in group II possesses valence number of +4.

Since XPS information comes from a depth less than tens of atom layers, so the XPS data in our experiments

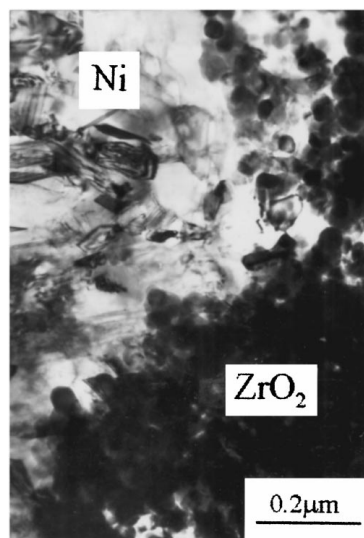


Figure 4 Interfacial structure between matrix Ni and ZrO₂ particles in Ni-ZrO₂ plating.

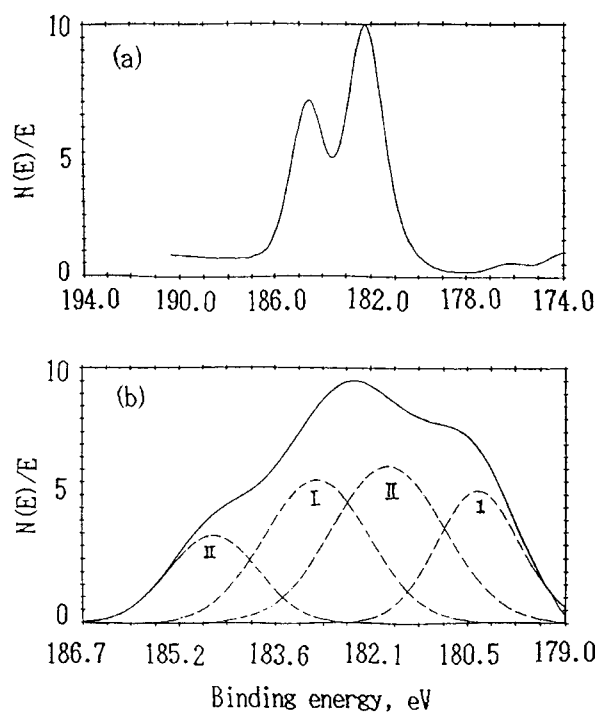


Figure 5 XPS spectra of the Zr3d region. —: tested curves ----: fitting curve a) ZrO₂ powder b) Ni-ZrO₂ plating.



Figure 3 Electron diffraction pattern of ZrO₂ particles.

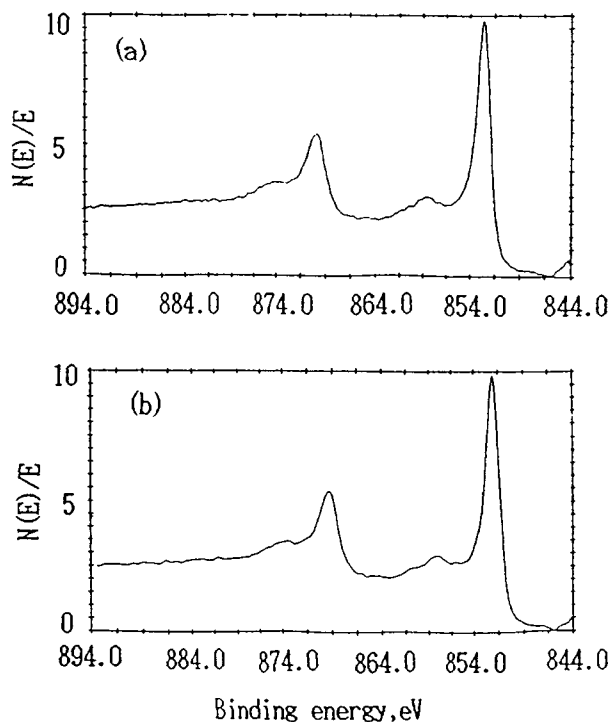


Figure 6 XPS spectra of the Ni2p region a) Ni plating b) Ni-ZrO₂ plating.

mainly reflect the state of the interfacial regions between matrix Ni and ZrO₂ particles. From the results, it can be concluded that Zr⁴⁺ on the surface of ZrO₂ particles may be partially reduced to Zr³⁺ during the process of CE, and the zirconium ions in the interfacial regions of Ni-ZrO₂ CP possess a different valence number such as +3, +4, or between +3 and +4.

A quantitative analysis has been done according to the O1s and Zr3d XPS peak areas of ZrO₂ powder that was not sputtered by Ar⁺ before the measurement. The result showed that the atomic ratio of Zr to O is 1 : 1.978, which is bigger than the value calculated from the chemical formula of ZrO₂. This indicates that the cavities of O²⁻ exist on the surface of ZrO₂ particles.

The XPS spectrum of Ni plating (Fig. 6a) shows the normal binding energy of 852.7 eV (Ni2P_{3/2}) and 870.0 eV (Ni2P_{1/2}). Comparing the two XPS spectra showed in Fig. 6, it can be found that the peak positions of Ni2P are different although their peak shapes are almost the same. The binding energies of Ni in CP are 852.1 eV (Ni2P_{3/2}) and 869.4 eV (Ni2P_{1/2}), and the binding energy of Ni2P_{3/2} electron has shifted by 0.6 eV towards the negative direction comparing with that of pure Ni plating. The resolution of the XPS measurements is within the limits of 0.2 eV, so the notable negative shift of the binding energy shows that a bonding action has been generated between matrix Ni and ZrO₂ particles, and Ni atoms prefer to obtain electrons during the bonding process. Since the chemical shift is less than 1 eV, this explains that the bond possesses a mixed property of covalent bond and ionic bond.

3.3. Results of UPS

UPS technique was used to investigate the difference in the electronic structures of nickel in Ni plating and CP.

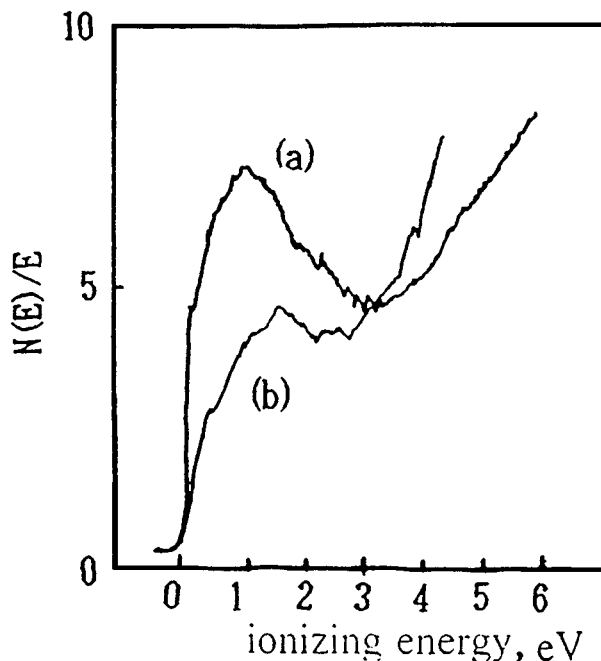


Figure 7 UPS spectra of Ni plating (a) and Ni-ZrO₂ plating(b).

The peaks near the Fermi level in Fig. 7 come from the d band of Ni, and the ionizing energies for Ni plating and CP are 1.0 eV and 1.6 eV, respectively. The 0.6 eV increase in the ionizing energy for CP means that the ionization of the d electrons of Ni is more difficult in CP than in Ni plating. This also proves that Ni atoms in CP participate in the bonding action. Fig. 7 also shows that the peak height of CP has clearly decreased comparing with that of Ni plating. This indicates that the emissive intensity of d electrons of Ni in CP has greatly decreased. The decrease in emissive intensity can also be taken as evidence that the d electrons of Ni atoms located at the interfacial regions participate in the bonding action with ZrO₂.

3.4. Results of ESR

As illustrated in Fig. 8, Ni produces an ESR spectrum and ZrO₂ powders do not produce an ESR spectrum. It was also found that the peak shapes and peak widths for Ni plating and CP are the same, but their peak heights are not equal. The calculated relative intensities of ESR peak for Ni plating and CP are 4.16/mg and 3.80/mg, respectively. The decrease in the ESR relative intensity of CP can be explained as the decrease of single electron numbers in matrix Ni, and the decreased numbers of electrons can be considered as those electrons which have participated in the bonding action. So the orbital interaction between matrix Ni and ZrO₂ particles was also verified by ESR.

3.5. Model of interfacial orbital interaction in CP

XPS results have proved that the valence number of zirconium in pure ZrO₂ powder is +4, but the valence number of zirconium in Ni-ZrO₂ CP has several different values such as +3, +4, even between +3 and +4.

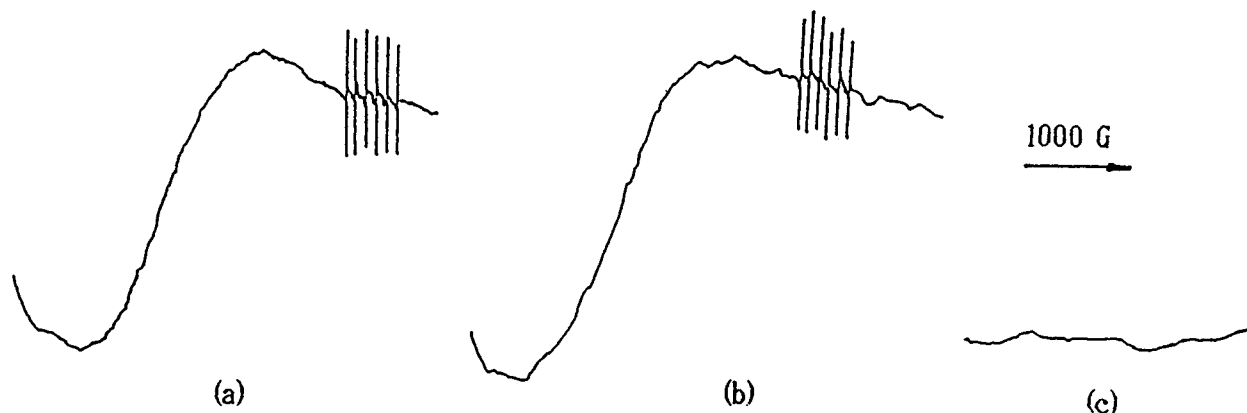


Figure 8 ESR spectra of Ni plating a), Ni-ZrO₂ plating b) and ZrO₂ powder c). (Mn²⁺ was used as standard).

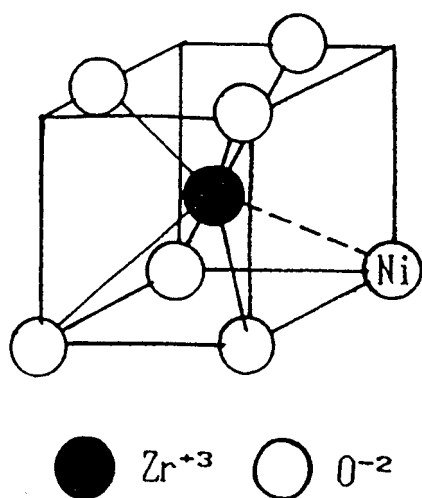


Figure 9 Schematic view of the bonding between Ni atoms and Zr³⁺ ion in Ni-ZrO₂ plating.

The differences indicate that the Zr⁴⁺ ions on the surface of ZrO₂ particles may be partially reduced to Zr³⁺ during the process of CE under high cathodic overpotential. The quantitative XPS analytical result of pure ZrO₂ powder has shown that O²⁻ cavities exist on the surface of ZrO₂ particles. All these make it possible that Ni atoms were preferentially absorbed to the O²⁻ cavities and approached to Zr³⁺ so sufficiently to bond with Zr³⁺ when Ni atoms contacted with ZrO₂ particles during the process of CE. The XPS results of CP have also shown that the binding energy of Ni 2P_{3/2} electrons has a negative chemical shift less than 1 eV comparing with that of Ni plating, and that zirconium ions with valence numbers between +3 and +4 have also been found. It can be concluded that during the bonding process, Zr³⁺ ions tend to give out electrons and Ni atoms tend to obtain electrons, and the bonds formed between them probably possess a mixed properties of covalent bond and ionic bond.

The UPS results have shown that it is the d orbit of Ni atoms that participates in the bonding process. Since the outer electronic structure of Zr atom is 4d² 5S², the bonding process can only take place among the d orbitals of Ni atoms and Zr³⁺ ions, and their d orbitals overlap mutually to bond. The molecular model of the interfacial orbital interaction in CP can be schematically shown in Fig. 9.

3.6. Effect of the orbital interaction between Ni and ZrO₂ particles on the electrocatalytic activity for HER

The electrocatalytic activity of transient metal for HER is greatly affected by the electron distribution in the outer d orbit of the atom. The fuller the d orbit is filled with electrons, the higher the electrocatalytic activity showed [8]. As indicated above, the d orbit of Ni atoms in the composite plating is more filled up with electrons through the orbital interaction between matrix Ni and ZrO₂ particles. The more filled up d orbit of Ni atoms would decrease the adsorptive heat of hydrogen atoms adsorbed on Ni during the process of HER, and makes the adsorbed hydrogen atoms desorb more easily. The HER mechanism of composite plating used for electrodes belongs to a recombination-desorption mechanism [4]. It is believed that the important reason for the high electrocatalytic activity of composite plating for HER is the more filled-up d orbit of Ni atoms caused by the orbital interaction between matrix Ni and ZrO₂ particles.

4. Conclusions

- 1) No new phases at the interfacial regions of matrix Ni and ZrO₂ particles in CP are found by either TEM observation or electron diffraction analysis.
- 2) XPS analyses indicate that Zr⁴⁺ ions on the surface of ZrO₂ particles were partially reduced to Zr³⁺ ions when ZrO₂ particles were embedded in Ni-ZrO₂ CP during the composite electrodeposition process under high cathodic over-potential.
- 3) XPS quantitative analysis shows that O²⁻ cavities exist on the surface of ZrO₂ particles.
- 4) The analyses of XPS, UPS and ESR have shown that the orbital interaction by the overlap of the d orbit between Ni atoms and Zr³⁺ ions exists at the interfacial regions in Ni-ZrO₂ CP.

References

1. R. J. YOUNG and X. YANG, *Composites* **27A**(9) (1996) 737.
2. D. ATTWOOD and P. I. MARSHALL, *Composites* **27A**(9) (1996) 775.
3. K. WOO and J. D. WHITCOMB, *J. Composite Materials* **30**(9) (1996) 984.
4. W. WANG, PhD dissertation, Tianjin University, 1992.

5. S. R. MOLISON and W. H. HUANG, "Electrochemistry of Semiconductors and Metal Oxides" (Science Press, Beijing, 1988).
6. W. WANG and H. T. GUO, *Material Research* **11**(2) (1997) 143.
7. C. D. WAGNER, W. M. RIGGS and L. E. DAVIS, "Handbook of X-Ray Photoelectron Spectroscopy" (Eden Prairie, Minnesota 55344, 1979).
8. ALLEN J BARD, "Encyclopedia of Electrochemistry of the Elements," Vol. IX, Part A (Marcel Dekker Inc., 1982).

*Received 14 October 1997
and accepted 24 June 1999*